

# Photoluminescence properties of $Y_2O_3$ co-doped with Eu and Bi compounds as red-emitting phosphor for white LED

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**Abstract** Bismuth doped  $Y_2O_3$ : Eu was used as a red phosphor with a very high efficiency and an appropriate emission wavelength of around 310–400 nm. This red phosphor was synthesized by the solid state reaction which is normally used in the field of white LEDs. In this study, we synthesized  $Y_2O_3$ : Eu, Bi phosphors using a solid state reaction. We investigated the effect of the  $Eu^{3+}$  and  $Bi^{3+}$  concentrations and additive fluxes on the emission characteristics. The fabricated phosphors were investigated by analyzing their particle size and crystal structure with scanning electron microscopy and X-ray diffraction (XRD). Their photoluminescence (PL) spectra were also measured at room temperature.

**Keywords** Red-emitting phosphor · White light emitting diode ·  $Y_2O_3$  co-doped Eu and Bi · Photoluminescence properties

## 1 Introduction

A three-band white-light-emitting diode (LED) consists of an LED device that emits either soft ultraviolet (UV) or blue light, and red-green-blue (RGB) phosphors [1–3]. For the development of the three-band white LEDs, InGaN and GaN chips have been both considered as excitation sources for the RGB phosphors. In this case, either organic dyes or inorganic phosphors can be used as the RGB components for the three band white LEDs [1–4]. However, the inorganic phosphors have some environmental problems during their preparation and use, because they include toxic elements, such as sulfur, chlorine, and cadmium [2–4]. To solve these problems, in the

present study, we attempted to develop new oxide for RGB phosphors, bismuth doped  $Y_2O_3$ : Eu, which was used as the red phosphor with a very high efficiency and an appropriate emission wavelength of around 340–400 nm. This red phosphor was synthesized by the solid state reaction with flux, which is normally used for the fabrication of white LEDs.

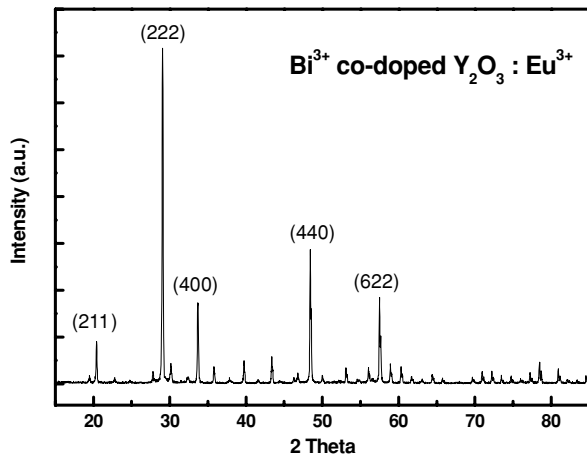
In this study, we investigated the effect of the  $Eu^{3+}$  and  $Bi^{3+}$  concentrations and additive fluxes on the emission characteristics of the fabricated phosphors, and the particle size and crystal structure were also investigated.

## 2 Experimental procedure

Bi doped  $Y_2O_3$ : Eu compounds were prepared by the conventional solid-state reaction. Eu and Bi ions were added in the form of  $Eu_2O_3$  and  $Bi_2O_3$ , respectively. Commercially available high purity reagents (99.99%) of  $Y_2O_3$ ,  $Eu_2O_3$ ,  $Bi_2O_3$ ,  $H_3BO_3$  and  $BaCl_2 \cdot 2H_2O$  were used for the production of the samples. In each case, the total weight of the powders was 10 g, and the powders were mixed together using an alumina mortar and pestle. The mixed batches were fired in an alumina crucible at 1100°C for 3 h in air. The fired specimens were ground using an alumina mortar and pestle, followed by washing and drying procedure. The dried powders were sieved to classify the powder size for the photoluminescence system measurements.

The emission and excitation spectra were measured at room temperature using a fluorescence spectrophotometer in the ranges of 550–700 nm and 250–550 nm, respectively. The crystalline phases of the synthesized samples were characterized by an X-ray diffractometer ( $CuK_{\alpha}$ , 30 KV, 100 mA, Rigaku) and the surface morphologies were observed using a scanning electron microscope (ESEM, XL-30, Philips).

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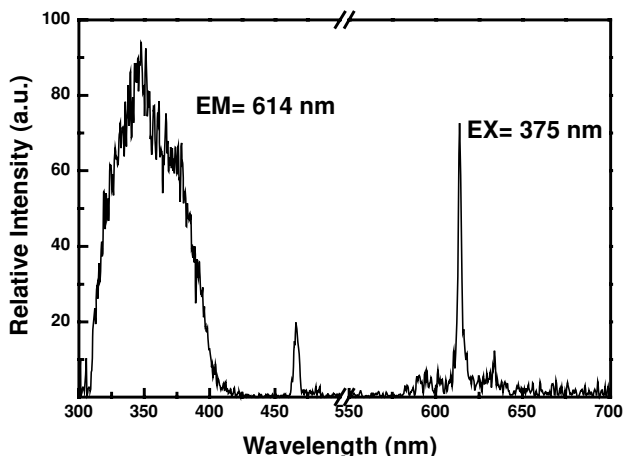


**Fig. 1** X-ray diffraction pattern of bismuth co-doped  $Y_2O_3:Eu$  fired at  $1100^\circ C$  for 3 h in air

### 3 Results and discussion

The XRD pattern of  $Y_2O_3$  sample co-doped with Eu and Bi is presented in Fig. 1. According to JCPDS card 41-1105, pure  $Y_2O_3$  has a cubic crystal structure with the  $Ia3(206)$  space group, and its lattice parameter is 1.641 nm. Only a single cubic phase was identified from XRD measurement without having any extraneous phase and the doped Eu and Bi ions had little influence on the host  $Y_2O_3$  structure. It was concluded that the co-doping with Eu and Bi increased the lattice parameters of the phosphor, due to the fact that the ionic radius of  $Y^{3+}$  (0.088 nm) is slightly lower than those of  $Eu^{3+}$  (0.109 nm) and  $Bi^{3+}$  (0.196 nm).  $Eu^{3+}$  and  $Bi^{3+}$  ions were expected to occupy the  $Y^{3+}$  sites in this phosphor. It is believed that this single phase was successfully developed through our preparation procedure.

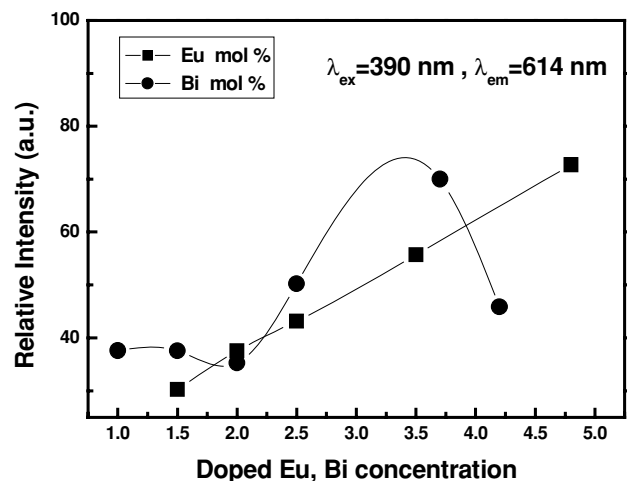
Figure 2 shows the emission and excitation spectra of  $Y_2O_3$  co-doped with  $Eu^{3+}$ ,  $Bi^{3+}$  measured at room temperature. The emission spectrum exhibited a narrow band



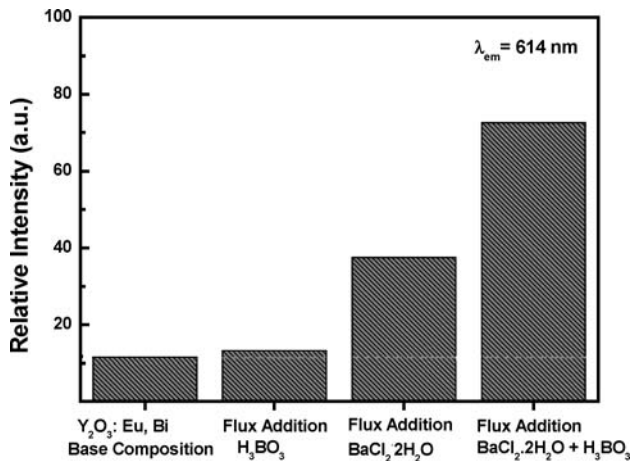
**Fig. 2** Emission ( $\lambda_{ex} = 375$  nm) and excitation ( $\lambda_{em} = 614$  nm) spectra of  $Y_2O_3:Eu^{3+}, Bi^{3+}$

(600–650 nm) having sharp peaks at about 610–620 nm due to  ${}^5D_0 \rightarrow {}^7F_2$  transition of  $Eu^{3+}$ . In addition, the excitation spectrum exhibited a broad band between 250 and 550 nm with peaks occurring about 310–400 nm and a sharp peak at 465.5 nm. Conventional  $Y_2O_3:Eu$  was not absorbed near UV efficiently.  $Y_2O_3:Bi$  exhibited absorption band at 375 nm and emission band around 410 nm because of  ${}^1S_0 \rightarrow {}^3P_1$  transition of  $Bi^{3+}$  [5–9]. When incorporated in  $Y_2O_3:Eu$  phosphors,  $Bi^{3+}$  ions act as a sensitizer for  $Eu^{3+}$  ions under 350 nm radiation. The energy transferring from  $Bi^{3+}$  ions to  $Eu^{3+}$  ions by no radiation process resulted in the final emission from excited  $Eu^{3+}$  ions.  $Bi^{3+}$  ion was introduced into the  $Y_2O_3$  red phosphors, excitation spectra were enhanced significantly and a new band located at 350–390 because of  $6s^2 \rightarrow 6s6p$  transition of  $Bi^{3+}$ . Figure 3 shows the emission spectra ( $\lambda_{ex} = 390$  nm,  $\lambda_{em} = 614$  nm) of the  $Eu^{3+}$  and  $Bi^{3+}$  co-doped yttrium oxide (with a fixed Eu concentration of 3.0 mol% and variable Bi concentration or vice versa) measured at room temperature as a function of the  $Eu^{3+}$  and  $Bi^{3+}$  concentrations.

With increasing  $Eu^{3+}$  concentration, the relative intensity of the red emission was continuously increased. However, with increasing  $Bi^{3+}$  concentration, the intensity of the red emission was increased and reached a maximum at a  $Bi^{3+}$  concentration of 3.5 mol %. Above this concentration, the intensity of the red emission decreased. Because of the energy transferred from the  $Eu^{3+}$  to the  $Bi^{3+}$  ions, the emission intensity of the  $Bi^{3+}$  ions increased, while that of the  $Eu^{3+}$  ions decreased in the  $Eu^{3+}$  and  $Bi^{3+}$  co-doped  $Y_2O_3$ . However, in the case where  $Y_2O_3:Eu$  was co-doped with Bi, all of the samples exhibited a strong absorption band in the near UV region because of the  ${}^1S_0 \rightarrow {}^3P_1$  transition of  $Bi^{3+}$ , while the strong 610–620 nm emission due to  $Eu^{3+}$  was retained.



**Fig. 3** Emission relative intensity ( $\lambda_{ex} = 390$  nm,  $\lambda_{em} = 614$  nm) of  $Eu^{3+}, Bi^{3+}$  co-doped yttrium oxide (with a fixed Eu concentration of 3.0 mol% and variable Bi concentration or vice versa) as a function of the  $Eu^{3+}$  and  $Bi^{3+}$  concentrations



**Fig. 4** Relative emission intensity of Y<sub>2</sub>O<sub>3</sub>:Eu<sup>3+</sup>, Bi<sup>3+</sup> at 614 nm with and without the H<sub>3</sub>BO<sub>3</sub> and BaCl<sub>2</sub>·2H<sub>2</sub>O fluxes

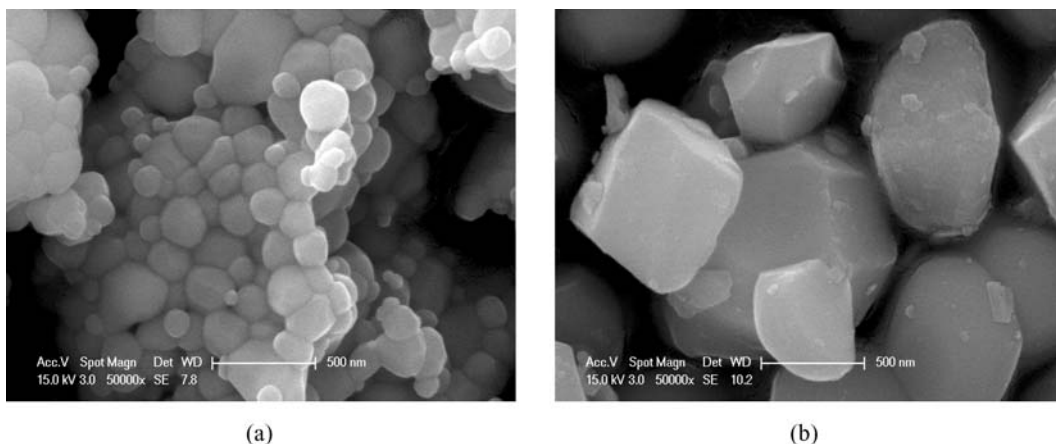
Figure 4 shows the relative emission intensity of Y<sub>2</sub>O<sub>3</sub>:Eu<sup>3+</sup>, Bi<sup>3+</sup> with or without H<sub>3</sub>BO<sub>3</sub> and BaCl<sub>2</sub>·2H<sub>2</sub>O flux at 614 nm. The relative emission intensity of Y<sub>2</sub>O<sub>3</sub>:Eu<sup>3+</sup>, Bi<sup>3+</sup> containing 0.43 mol% H<sub>3</sub>BO<sub>3</sub> was slightly higher than that of the base composition sample. The relative emission intensity of Bi<sup>3+</sup>-co-doped Y<sub>2</sub>O<sub>3</sub>:Eu<sup>3+</sup>, containing 2.08 mol% BaCl<sub>2</sub>·2H<sub>2</sub>O was approximately 3.3 times higher than that of the base composition sample. Furthermore, H<sub>3</sub>BO<sub>3</sub> was added to the Bi<sup>3+</sup>-co-doped Y<sub>2</sub>O<sub>3</sub>:Eu<sup>3+</sup> 2.08 mol% BaCl<sub>2</sub>·2H<sub>2</sub>O sample in order to improve the emission intensity. The relative emission intensity of Y<sub>2</sub>O<sub>3</sub>:Eu<sup>3+</sup>, Bi<sup>3+</sup> containing 0.43 mol% H<sub>3</sub>BO<sub>3</sub> and 2.08 mol% BaCl<sub>2</sub>·2H<sub>2</sub>O was approximately 6.4 times higher than that of the base composition sample. This improvement in the emission intensity associated with the Eu<sup>3+</sup> and Bi<sup>3+</sup> ions in the Y<sub>2</sub>O<sub>3</sub> samples was attributed to the synergetic effect afforded by the H<sub>3</sub>BO<sub>3</sub> and BaCl<sub>2</sub>·2H<sub>2</sub>O fluxes.

Figure 5(a) shows the SEM image of the Y<sub>2</sub>O<sub>3</sub>:Eu<sup>3+</sup>, Bi<sup>3+</sup> base composition samples. Spherically shaped crystals with

diameters of less than 300 nm grew and became hard and aggregated, having to be partially broken up again in a mortar. The samples were uniform without a visible admixture of any impurity phases. Figure 5(b) shows the SEM image of the sample containing 0.43 mol% H<sub>3</sub>BO<sub>3</sub> and 2.08 mol% BaCl<sub>2</sub>·2H<sub>2</sub>O. The phosphor particles grew to achieve diameters of 400–800 nm. Also, the Y<sub>2</sub>O<sub>3</sub>:Eu<sup>3+</sup>, Bi<sup>3+</sup> phosphor particles containing the fluxes exhibited no aggregation and had regular morphology characteristics without any visible admixture of any impurity phases. It was observed that the small sized Bi<sup>3+</sup> co-doped Y<sub>2</sub>O<sub>3</sub>:Eu<sup>3+</sup> particles exhibited poor luminescence intensity, but that when the particle size grew due to the addition of the fluxes, the emission intensity increased by approximately 6.4 times.

#### 4 Conclusion

The red emission properties of Bi<sup>3+</sup> co-doped Y<sub>2</sub>O<sub>3</sub>:Eu<sup>3+</sup> prepared by the solid-state reaction were investigated, in order to verify its potential to act as the red emitting phosphor of white LEDs. The emission spectrum consisted of a narrow band at 600–650 nm, with sharp peaks occurring at about 610–620 nm due to the <sup>5</sup>D<sub>0</sub> → <sup>7</sup>F<sub>2</sub> transition of Eu<sup>3+</sup>, while the excitation spectrum exhibited a broad band between 250 and 550 nm with peaks occurring in the range of 310–400 nm and a sharp peak being observed at 465.5 nm. In the emission spectra (λ<sub>ex</sub> = 390 nm) of Eu<sup>3+</sup>, Bi<sup>3+</sup>-doped yttrium oxide with a fixed Bi concentration of 3.0 mol% and increasing Eu<sup>3+</sup> concentration, the relative intensity of the red emission increased continuously. However, in the emission spectra of the Eu<sup>3+</sup>, Bi<sup>3+</sup>-doped yttrium oxide with a fixed Eu concentration of 3.0 mol% and increasing Bi<sup>3+</sup> concentration, the intensity of the red emission increased and reached a maximum at a Bi<sup>3+</sup> concentration of 3.5 mol%. Above this concentration, the intensity of the red emission decreased. Also, the relative emission intensity of



**Fig. 5** SEM image of (a) Eu<sup>3+</sup>, Bi<sup>3+</sup> co-doped yttrium oxide and (b) Eu<sup>3+</sup>, Bi<sup>3+</sup> co-doped yttrium oxide containing 0.43 mol% H<sub>3</sub>BO<sub>3</sub> and 2.08 mol% BaCl<sub>2</sub>·2H<sub>2</sub>O

the  $\text{Y}_2\text{O}_3:\text{Eu}^{3+}, \text{Bi}^{3+}$  sample containing 0.43 mol%  $\text{H}_3\text{BO}_3$  and 2.08 mol%  $\text{BaCl}_2 \cdot 2\text{H}_2\text{O}$  was approximately 6.4 times higher than that of the base composition.

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